Acta Crystallographica Section C Crystal Structure Communications

ISSN 0108-2701

Cerium iron sulfide, $Ce₃Fe_{1.94}S₇$

Allison M. Mills and Michael Ruck*

Institut für Anorganische Chemie, Technische Universität Dresden, D-01062 Dresden, Germany Correspondence e-mail: michael.ruck@chemie.tu-dresden.de

Received 13 April 2004 Accepted 7 June 2004 Online 10 July 2004

Tricerium(III) diiron(II,III) heptasulfide, $Ce₃Fe_{1.94}S₇$, crystallizes in the polar hexagonal space group $P6₃$ and adopts the $Ce₆Al_{3.33}S₁₄$ structure type. The Fe atoms occupy both tetrahedral and octahedral sites. Isolated FeS₄ tetrahedra, all pointing in the same direction, are stacked along the threefold rotation axes. Chains of face-sharing $FeS₆$ octahedra propagate along the $6₃$ axis. Vacancies resulting from the partial oxidation of Fe^{2+} to Fe^{3+} occur exclusively in the octahedral Fe sites. The Ce atoms are coordinated by [7+1] S atoms, which form bicapped trigonal prisms.

Comment

In an early investigation of the pseudo-binary systems R_2S_3 MS (R is La–Nd and M is Mn–Ni), a series of compounds originally described with the formula $R₄MS₇$ were discovered (Collin et al., 1968). The cell parameters of these compounds, determined from X-ray powder diffraction patterns indexed in the hexagonal space group $P6_3$, were reported (for 'Ce₄FeS₇', $a = 10.202$ Å and $c = 5.657$ Å). Since then, more detailed structural information has been lacking for these compounds. We report here the single-crystal structure determination of one member of the series, which has the actual formula $Ce₃Fe_{1.94}S₇.$

Figure 1

A view of the structure of $Ce_3Fe_{1.94}S_7$ along [001]. Displacement ellipsoids are drawn at the 95% probability level.

Figure 2

A view along $[100]$ of part of the structure of $Ce₃Fe_{1.94}S₇$, showing the stacked FeS₄ tetrahedra and the FeS_{6/2} chains of face-sharing octahedra, which are connected by CeS_{7+1} bicapped trigonal prisms. Displacement ellipsoids are drawn at the 95% probability level. [Symmetry codes: (i) $1-x, 1-y, z - \frac{1}{2}$; (vii) $x - y, x, z - \frac{1}{2}$.]

The title compound adopts the $Ce₆Al_{3.33}S₁₄$ structure type (de Saint-Giniez et al., 1968), which is also sometimes referred to as the La₃CuSiS₇ structure type (Flahaut & Laruelle, 1970). Thus, $Ce_3Fe_{1.94}S_7$ belongs to a large family of compounds of general formula R_3 *MM'* Q_7 , where *R* is a rare-earth metal, *M* and M' are metals or metalloids, and O is a chalcogen (Villars, 1997; Flahaut & Laruelle, 1970). Some other recently reported $R_3MM'Q_7$ compounds include Y₃NaSiS₇ (Hartenbach & Schleid, 2003), La₃CuGe Q_7 (Q is S or Se; Poduska *et al.*, 2002) and $La₃Al_{0.44}Si_{0.93}S₇$ (Yang & Ibers, 2000).

Views of the $Ce₃Fe_{1.94}S₇$ structure, highlighting the Fecentred coordination polyhedra, are presented in Figs. 1 and 2. The Fe atoms occupy two types of sites, with tetrahedral and octahedral geometries, respectively. Isolated $Fe1S₄$ tetrahedra, all pointing in the polar [001] direction, are stacked along the threefold rotation axes (Fig. 2). The tetrahedra are trigonally compressed, with one shorter Fe1 $-$ S3 distance of 2.224 (6) \AA and three longer Fe1 $-S2$ distances of 2.265 (3) Å (Table 1). These distances are somewhat longer than those found in the FeS₄ tetrahedra of La₃MnFeS₇ (2.11–2.22 A; Nanjundaswamy & Gopalakrishnan, 1983), but shorter than those found in the tetrahedra of La₂Fe₂S₅ (2.30–2.37 Å; Besrest & Collin, 1977). The S-Fe1-S angles of 106.67 (12) and 112.14 (11) \degree are close to the ideal tetrahedral value of 109.5°.

The Fe2S₆ octahedra share opposite faces to form linear $FeS_{6/2}$ chains that propagate along the 6_3 axis (Fig. 2). The Fe atoms within the slightly elongated $Fe2S1₆$ octahedra are shifted closer to one shared face than the other, yielding Fe2 $-$ S1 distances of $2.532(4)$ and $2.578(4)$ Å (Table 1). These distances are similar to those observed in the more distorted FeS₆ octahedra of Ce₂Fe_{1.82}S₅ (2.466–2.702 Å; Harms *et al.*, 2004). The cis -S $-Fe$ 2 $-$ S angles range from 88.40 (15) to 90.58 (5) \degree and the *trans* angles are 178.6 (2) \degree .

As in the $Ce₆Al_{3.33}S₁₄$ structure (de Saint-Giniez et al., 1968), vacancies occur in the octahedral site of $Ce₃Fe_{1.94}S₇$ [Fe2, with a refined occupancy of 0.942 (16)]. The defects in $Ce₃Fe_{1.94}S₇$ are presumed to result from the partial oxidation of Fe^{2+} to Fe^{3+} , according to the charge-balanced formula

 $(Ce^{3+})_3(Fe^{3+})_{1,12}(Fe^{2+})_{0.82}\square_{0.06}(S^{2-})_7$. In support of this hypothesis, a recent Mößbauer investigation of the nonstoichiometric cerium iron sulfide $Ce₂Fe_{1.82}S₅$ established that the oxidation of Fe^{2+} to Fe^{3+} accompanies the formation of vacancies in this compound (Harms et al., 2004).

The Ce atoms, located between the Fe-centred polyhedra, are coordinated by [7+1] S atoms: one S3, two S2, and three S1 atoms, at distances of 2.843 (3)–3.037 (3) \AA , form a trigonal prism, capped by one S2 atom at a distance of 3.000 (3) \AA and one S1 atom at a significantly longer distance of 3.414 (4) \AA (Fig. 2 and Table 1). The shorter distances are within the range observed in the CeS₈ bicapped trigonal prisms of Ce₂Fe_{1.82}S₅ $(2.887-3.142 \text{ Å})$; Harms *et al.*, 2004). A similar but more pronounced [7+1] coordination is observed in $Ce₆Al_{3,33}S₁₄$, where the corresponding short and long $Ce-S$ distances are 2.83–3.03 and 3.58 (1) \AA , respectively (de Saint-Giniez *et al.*, 1968).

Experimental

Hexagonal prisms of $Ce_3Fe_{1.94}S_7$ were isolated from the reaction of the elements in an alkali chloride flux. The starting reagents were cerium (rod, 99.85%, Treibacher; freshly filed prior to use), iron (powder, 99.99%, ABCR), and sulfur [powder, >99%, VEB Laborchemie; recrystallized from CS_2 , then purified of C according to the method of von Wartenberg (1956)]. A 1:1 mixture of LiCl (p.a., Merck) and KCl (p.a., J. T. Baker) was used as a flux after being heated under dynamic vacuum to remove any moisture. The elements, in a ratio of 3Ce:2Fe:7S (0.25 g in total), were added to the LiCl–KCl flux (0.5 g) in a fused silica ampoule $(6 \text{ cm in length}, 0.8 \text{ cm})$ in diameter), which was then sealed under vacuum $(10^{-3}$ Torr; 1 Torr $= 133.322$ Pa). The reaction mixture was heated at 1170 K for 4 d and then cooled to room temperature at a rate of 10 K h^{-1} . The flux was removed by washing the sample several times with water and ethanol. The major component of the product was $Ce₃Fe_{1.94}S₇$. Energydispersive X-ray (EDX) analysis confirmed the presence of Ce, Fe, and S; analysis (mol. %): Ce 24 (1), Fe 14 (1), S 62 (1).

Crystal data

Table 1

Selected geometric parameters (\mathring{A}, \degree) .

Symmetry codes: (i) $1 - x$, $1 - y$, $z = \frac{1}{2}$; (ii) $-x + y$, $-z$, z; (iii) $1 - x + y$, $1 - x$, z; (iv) $1-x, -y, z-\frac{1}{2}$; (v) $y, -x+y, z-\frac{1}{2}$; (vi) $y, -x+y, \frac{1}{2}+z$; (vii) $x-y, x, z-\frac{1}{2}$; (viii) $-x, -y, z - \frac{1}{2}$.

Data collection

 w

 \overline{S}

39 parameters $w = 1/[\sigma^2 (F_o^2) + (0.0988P)^2]$ $+ 5.9571P$ where $P = (F_o^2 + 2F_c^2)/3$

 $(\Delta/\sigma)_{\text{max}} < 0.001$ $\rm{\AA}^{-3}$ e Å $^{\rm -3}$ rection: SHELXL97 Extinction coefficient: $0.035(3)$ Absolute structure: Flack (1983), 390 Friedel pairs

Flack parameter = $0.02(6)$

Since the displacement ellipsoid of the Fe2 octahedral site was initially unusually large, the occupancy of the site was refined freely, resulting in a value of 0.942 (16). Refinement of the occupancy of the Fe1 tetrahedral site confirmed that it is fully occupied. Atomic positions were standardized using the program STRUCTURE TIDY (Gelato $& Parthé, 1987$).

Data collection: *IPDS* (Stoe & Cie, 2000); cell refinement: *IPDS*; data reduction: *IPDS*; program(s) used to solve structure: $SHELX$ S97 (Sheldrick, 1997); program(s) used to refine structure: SHELXL97 (Sheldrick, 1997); molecular graphics: DIAMOND (Brandenburg, 1999).

The authors gratefully acknowledge the financial support of the Deutsche Forschungsgemeinschaft (DFG) within the SFB 463.

Supplementary data for this paper are available from the IUCr electronic archives (Reference: BC1047). Services for accessing these data are described at the back of the journal.

References

- Besrest, F. & Collin, G. (1977). J. Solid State Chem. 21, 161-170.
- Brandenburg, K. (1999). DIAMOND. Release 2.1c. Crystal Impact GbR, Bonn, Germany.

Collin, G., Rouyer, F. & Loriers, J. (1968). C. R. Acad. Sci. Ser. C, 266, 689-691. Flack, H. D. (1983). Acta Cryst. A39, 876-881.

- Flahaut, J. & Laruelle, P. (1970). The Chemistry of Extended Defects in Non-Metallic Solids, edited by L. Eyring & M. O'Keeffe, pp. 109-123. Amsterdam: North-Holland.
- Gelato, L. M. & Parthé, E. (1987). J. Appl. Cryst. 20, 139-143.
- Harms, W., Mills, A. M., Söhnel, T., Laubschat, C., Wagner, F. E. & Ruck, M. (2004). Solid State Sci. Submitted.

Hartenbach, I. & Schleid, T. (2003). J. Solid State Chem. 171, 382-386.

- Nanjundaswamy, K. S. & Gopalakrishnan, J. (1983). J. Solid State Chem. 49, 51±58.
- Poduska, K. M., DiSalvo, F. J., Min, K. & Halasyamani, P. S. (2002). J. Alloys Compd, 335, L5-L9.
- Saint-Giniez, D. de, Laruelle, P. & Flahaut, J. (1968). C. R. Acad. Sci. Ser. C, 267, 1029±1032.
- Sheldrick, G. M. (1997). SHELXS97 and SHELXL97. University of Göttingen, Germany.

Stoe & Cie (1999). X-SHAPE. Version 1.06. Stoe & Cie, Darmstadt, Germany.

Stoe & Cie (2000). IPDS. Version 2.93. Stoe & Cie, Darmstadt, Germany. Stoe & Cie (2001). X-RED. Version 1.22. Stoe & Cie, Darmstadt, Germany.

Villars, P. (1997). Pearson's Handbook, Desk Edition. Materials Park, Ohio: ASM International.

Wartenberg, H. von (1956). Z. Anorg. Allg. Chem. 286, 243-246.

Yang, Y. & Ibers, J. A. (2000). J. Solid State Chem. 155, 433-440.